Vol. 75, No. 5

AUGUST, 1954

Properties of Lactoprene BN-12.5, a Copolymer of Butyl Acrylate and Acrylonitrile, and Its Vulcanizates

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The knowledge gained in the development of Lactoprene EV, a copolymer of ethyl acrylate and chloroethyl vinyl ether, was used in developing another synthetic rubber designated Lactoprene BN, a copolymer of butyl acrylate and acrylonitrile. It may be prepared by the conventional polymerization techniques to yield a tack-free white product. The rubber can be handled on the ordinary rubber processing equipment and it bands readily when the rolls of the rubber mill are maintained at 150°F. By the proper choice of curatives, the vulcani-

zates exhibit outstanding resistance to heat and hot lubricating oils. The water resistance of the vulcanizates is significantly improved over that of Lactoprene EV, and the brittle points are substantially lower. This paper describes the properties of the vulcanizates obtained from a copolymer prepared from a monomer mixture of 87½% butyl acrylate and 12½% acrylonitrile. The properties of several commercially produced synthetic rubbers are included in this paper for purposes of comparison.

SEVERAL years ago this Laboratory published several papers (1-8) describing methods of preparing polymers and copolymers of ethyl acrylate which could be vulcanized with curatives frequently employed for diene rubbers. These early papers also summarized the chemical and physical properties of the vulcanizates. The most promising copolymer, ethyl acrylate and chloroethyl vinyl ether, designated Lactoprene EV, is now commercially available as Hycar PA-21.

The early studies demonstrated that conventional compounding facilities can be employed to incorporate the reinforcing agents and curatives. It was shown that several curatives and accelerators commonly used in the with advantage with the saturated acrylic rubbers. However, an amine, Trimene Base, and sulfur recipe was found to impart outstanding resistance to dry heat. In addition, the low volume swelling of these vulcanizates in lubricating oils suggests many applications such as O-rings, gaskets, oil seals and similar mechanical goods. These vulcanizates of Lactoprene EV exhibit certain

vulcanization of natural rubber and GR-S can be used

These vulcanizates of Lactoprene EV exhibit certain limitations. For example, the brittle point, about +12°F., is relatively high and the swelling in water at 100°C., about 50% in 48 hours, is excessive. In an effort to correct these deficiencies the copolymers of butyl acrylate and acrylonitrile were investigated. A previous paper (9) summarized the preliminary data on several butyl acrylate-acrylonitrile compositions. The present paper describes the recently completed evaluation

Note: The Eastern Regional Research Laboratory is one of the laboratories of the Bureau of Agricultural and Industrial Chemistry, Agricultural Research Administration, United States Department of Agriculture.

Table I — Osmotic Pressure Molecular Weights and Viscosities of the Lactoprene BN-12.5 Copolymers

Sample No.	Molecular Weight	Intrinsic Viscosity a,b
19 Q 1B1 ° T-302 d T-306 d	1,760,000 901,000 780,000	5.46 3.38

(a) Acetone used as solvent. (b) Concentration in g./100 ml. (c) Prepared in Government Laboratories at the University of Akron. (d) Prepared in the laboratory.

of the copolymer prepared from a monomer composition of $87\frac{1}{2}\%$ butyl acrylate and $12\frac{1}{2}\%$ acrylonitrile. This material has been designated Lactoprene BN-12.5 in accordance with the practice of the acrylic rubber development of this Laboratory. The number following the name indicates the amount of acrylonitrile in the monomer mixture.

Preparation and Characterization of the Raw Rubber

The procedure and the recipe employed for preparing these copolymers is substantially the same as that described in the previous publication (9). A paper describing in greater detail the preparation of acrylic ester-acrylonitrile copolymers has recently been published (10). The material used in this evaluation study was part of a pilot plant batch prepared in the Government Laboratories of the Office of Synthetic Rubber operated by the University of Akron. The recipe employed for this material was similar to those used in laboratory batches; however, a substantial amount of prefloc formed in the large reactor which did not form in the laboratory-prepared samples.

Acrylic monomers, because of their reactive double, bond, polymerize readily into high molecular weight polymers. Frequently, insoluble products are formed which nevertheless can be compounded in the conventional manner. The solution polymerization, using benzene in moderate concentrations as a solvent, always yields a soluble polymer with high molecular weight.

The emulsion method for the polymerization process frequently results in a polymer containing a low percentage of insoluble gel. Table I illustrates the osmotic pressure molecular weights and intrinsic viscosities for several polymers.

Compounding and Curing

Since these polymers are substantially saturated, the vulcanization process may not be the same as for the diene type rubbers. However, many curatives used in the vulcanization of these latter materials are also operative with acrylic rubbers. The experience gained in the development of Lactoprene EV indicated that the acrylic rubbers cured with an amine and sulfur should have good resistance to dry heat. These conclusions were substantiated in this study and Table II illustrates the recipes used in the evaluation of Lactoprene BN-12.5.

As with Lactoprene EV but in contrast to many diene polymers no initial breakdown is required. Furthermore, these acrylic copolymers band readily on the mill. There is a tendency for the materials to adhere to the rubber compounding rolls prior to the addition of the carbon black. In order to reduce the tendency for the stock to stick to the back roll, stearic acid is incorporated. Usually one part of stearic acid suffices and the rubber then remains on the front roll during compounding. In this study, however, we have employed an internal mixer for incorporating the filler and stearic acid. This is convenient since premastication is not necessary and the mixing schedule is shortened.

The following compounding scheme was found to be satisfactory for Lactoprene BN-12.5: The raw rubber is placed in the Banbury mixer maintained at 180°F. It is run at low speed (95 r.p.m. on the "Midget") for about one-half minute. The carbon black and stearic acid which have been superficially mixed are then added and the mixing continued for an additional two minutes at low speed. Mixing is continued for another three minutes at high speed (190 r.p.m. on the "Midget"). The compounded stock may tend to adhere to the mixer blades if the temperature rises too high. However, it can be readily removed by

TABLE II—THE EFFECT OF VARIOUS RATIOS OF TRIETHYLENE TETRAMINE AND SULFUR ON THE TENSILE PROPERTIES OF LACTOPRENE BN-12.5 VULCANIZATES CONTAINING FURNEX CARBON BLACK

	Α	В	С	D	Τ.	-	_					MILA	CAR	BON 1	OLACE		
Lactoprene BN-12.5 Stearic acid Semi-reinforcing furnace	100 1	100 1	100		E 100 1	F 100 1	G 100 1	H 100 1	1 100 1	J 100 I	K 100 1	100 1	M 100 1	N 100	O 100	P 100	Q 100
black (Furnex) Sulfur Triethylene tetramine	50 0 0.75		50 0.5 0.75	50 0.75 0.75	50 0 1	50 0.25 1	50 0.5 1	50 0.75 1	50 1 0.5	50 1 1	50 1 1.5	50 1	50 2	50	50 4	50 5	50
Tonoile et a .			In	itial P	ropert	ies Cu	red 60	Minute	s at 1	208°E	1.5		1	1	1	1	2
Tensile strength, p.s.i	790	1350	1380	1380	980	1310	1460	1490	N	1400	1370	1290	1360	1400	1370	1380	1390
Ultimate elongation, %	780	480	480	400	650											2000	1000
Shore A hardness (30 cm)	36	44		490	650	310	300	330	C	330	230	120	270	290	290	300	420
Modulus at 100%	90 230	160 390	48 180 480	46 190 480	40 130 290	54 260 740	55 310 860	55 330 830	Ř	55 250 720	60 420 1200	71 990	59 350 940	60 380	59 380	60 400	46 180
Tensile strongel					Aged	72 H	ours at	350°F.		-			240	950	930	930	420
Tensile strength, p.s.i Ultimate elongation, % Shore A hardness (30 sec.) Modulus at 100%	320 130 60 250	1200 250 64 390	1380 240 62 490	1450 250 62 450	460 130 61 360	1160 160 68 640	1390 140 76 1000	1390 - 140 - 76 - 940 -		1300 130 73 1020	1390 90 80	1650 50 91	1320 100 80 1300	1390 100 80 1390	1300 100 79 1300	1350 110 80 1330	710 70 75

cooling. In general, the samples reach a temperature of about 300°F, at the end of this mixing cycle. The curing ingredients are then incorporated on the conventional rubber compounding rolls which are maintained at 150°F.

Advantage in Amine-Sulfur Recipe

The development of Lactoprene EV demonstrated the advantage gained in the improvement of resistance to dry heat using an amine-sulfur recipe. The preliminary evaluation of the acrylate-acrylonitrile copolymers (9) established the need of sulfur for a heatresistant amine recipe. The sulfur is invariably added first to the carbon black-loaded stocks following a two minute milling cycle. The sulfur is usually dispersed in about two minutes. Finally triethylene tetramine is added. There is a tendency for the stock to split and stick to both rolls if the amine is added too quickly. However, once the triethylene tetramine is completely dispersed the stock remains on the front roll. The addition of the amine is usually completed in two minutes and the stock is then refined by closing the rolls The rolls and passing it through about six times. are then opened and the stock sheeted out to the proper thickness and allowed to cool. It should be kept in mind that this milling scheme has proved satisfactory in this study and is similar in many respects to the procedure developed for Lactoprene EV; it may not be the best industrially.

Satisfactory cures may be obtained in 15 to 60 minutes at 298°F. using the typical pressure molds as in A.S.T.M. Designation D15-41. The samples may also be vulcanized at 350°F, without appreciably affecting the final properties of the vulcanizates. Since the vulcanizates tend to adhere to the mold surface it is customary to use a mold release agent. Dow-Corning Mold Release No. 35*, when diluted 100 parts of water to one part of the mold release agent, has been found very effective. The diluted solution is brushed on the surfaces of the preheated mold and the excess wiped off lightly with a cloth. With this technique the cured test slabs were readily removed from the mold. In this evaluation study the compounded samples were vulcanized about 2 hours after compounding. The effect of storage of the compounded stocks at room temperature will be discussed.

Development of Test Recipes

Acrylic rubbers may be classed as specialty rubbers which are utilized in industrial applications where specific properties are required. Since these acrylic vulcanizates are apparently applicable where resistance to heat is of prime importance, this evaluation study was restricted to the determination of heat-resistant vulcanization recipes.

In order to facilitate the screening of the resistance to heat of the acrylic vulcanizates the tensile properties of the samples aged 72 hours in an air convection oven were obtained. Those stocks which did not suffer any significant change in their tensile strength compared with the initial tensile strength and still exhibited elongations of 100% or more were considered satisfactory and the vulcanizates were then given a more complete evaluation. The tensile tests obtained in this study were conducted according to ASTM specifi-

^{*} Mention of commercial names in this paper does not imply endorsement by the Department of Agriculture over similar products not named.

Tensile strength, p.s.i. Ultimate elongation, %		Tensile strength, p.s.i. Ultimate elongation, % Shore A hardness (30 sec.) Modulus at 100%		Lactoprene BN-12.5 Stearic acid	Table III—The Effect of Various Ratios of Triethylene Tetramine and Sulfur on the Containing Philblack O
980 1 320 60 400		1210 1 540 . 55 290 590		A 100 1 50 0 0.5	6 F 7
1430 1 460 64 410		1180 1 560 56 250 270		100 100 50 0.5	CRIET
1450 500 65 430		1130 650 54 260 570		C 100 1 50 1.0 0.5	HYLE
1450 450 65 430		1100 670 55 250 500	Initial	D 100 1 50 2 0.5	NE T
810 140 74 610	Ag	1490 450 56 280 590	l Prop	100 100 50 0 1.0	ETRAI
1550 140 81 1210	ed 72	1710 270 69 520 1200	erties	F 100 1 50 0.5 1.0	AINE
1710 140 84 1240	Hour	1690 270 68 510 1190	Cured	G 100 1 50 1.0	AND
1910 150 84 1230	Aged 72 Hours at 350°F	1830 300 69 520 1200	at 60	H 100 1 50 2 1.0	SULF
710 80 81	50°F.	1530 420 58 260 600	Min. a	I 100 50 0 1.5	ETRAMINE AND SULFUR ON CONTAINING PHILBLACK O
1710 80 90		1660 180 77 870	Initial Properties Cured at 60 Min. at 298°F.	J 100 50 0.5	THE
1760 70 90		1720 170 77 970	.11	100 100 100 100 1.0	TENSILE
1870 70 91		1790 180 76 940		100 100 50 2 1.5	SILE
ij		1450 450 55 210 490		M 100 100 50 0 2	Propi
O BE TO T		1440 1110 80 1290		100 100 50 0.5	ERTIES
TO BRITTLE TO TEST		1660 1110 81 1420		O 100 1 50 1.0 2	of I
Ħ		1780 130 83 1450		P 100 50 2 2	ACTO
1560 300 72 560		1530 450 60 400 770		100 100 50 0.5 0.75	PRENE
1650 270 74 620		1650 440 60 400 820		R 100 1 50 1.0 0.75	BN-
1590 290 75 570		1520 500 60 390 750		S 100 100 50 0.75	12.5 \
1620 250 77 630		1630 410 61 450 870		T 100 1 50 3 0.75	Properties of Lactoprene BN-12.5 Vulcanizates
1790 160 85 1180		1920 380 65 490 1000		100 50 1	NIZA
1780 170 87 1160		1790 370 65 480 990		100 50 1	TES

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Tarle IV—The Effect of Various Ratios of Triethylene Tetramine and Sulfur on the Tensile Properties of Lactoprene BN-12.5 Vulcanizates Containing P-33 and Wyex Carbon Blacks

TARLE IV—THE EFFECT OF V TIES OF LACTOPRENE	ARIO BN-1	us Ka 12.5 \	ULCA	NIZAT	res C	ONTAI	NING	P-33	3 ANI) VV ¥ Т	K C	L	M	N	0	P
TIES OF LACTOR REAL	A	В	C	ע	E 100	F 100	G 100	H 100	100	100	100	100	100 1	100 1	100 1	100 1
Lactoprene BN-12.5	100	100	100	100 1 100	1 100	1 100	1 100	1 100								50
Stearic acid	100	100	100	100			<u> </u>		50 1.0	50 1.0	50 1.0	50 1.0 2.5	50 0	50 0.5	50 2	3 2
Easy processing channel black— Wyex Sulfur	1.0	1.0 1.0	1.0 1.5	1.0	0 1.0	0.5 1.0	1.0	1.0	1.0	1.5	2	2.5	. 2		-	
Sulfur Triethylene tetramine	0.5	1.0 I1	nitial I	roper	ies Cu	red 60	Min.	at 29	8°F.	1250	1480	1580	1200	1220	1420	1480 260
uth psi	Ņ	1520	1640 310	1550 190	1070 630	1540 340	1380 370	1460 390 59	940 700 50	450 60	300 70	240 73	460 56	420 64	240 68	68
Tensile strength, p.s.i	Ċ.	440 53	63	71	45	60	59	290	280	430	630	810	410	490 780	780 1290	780 1290
as 1 1 as at 100%	R	220 590	390 980	720	170 310	370 800	280 620	660		790	1140	1390	710	760	12,0	
200%	Ľ	370		Age	d 72 H	ours a				1050	1300	1390	670		1380 90	
Tensile strength, p.s.i.		1190				1140 120		150	220	170	110	80 90	110 72	80	86	
Ultimate elongation, /		- 190 - 70 - 610	82	91	68 460								670	860		
Modulus at 100%		- 010														

cations D-412-41 using a die similar to Die D except that the reduced section was 1/4-inch wide.

Tables II, III and IV show the effect of various ratios of sulfur and triethylene tetramine on the initial tensile properties with four carbon black fillers—Furnex, Philblack O, P-33 and Wyex. The carbon black loadings used in this study were restricted to those illustrated in the tables. A previous inspection showed that these carbon black concentrations gave adequate reinforcement; however, other carbon black loadings may result in stocks superior to those described in this paper.

These tables also show the change in properties on heat aging in an air convection oven. As can be seen, it is possible to obtain satisfactory vulcanization in the absence of sulfur but on exposure to temperatures of 350°F, there is an inordinate decrease in the tensile strength of the vulcanization regardless of the type of carbon black filler used. See for example Table II recipes A, E and Q; Table III recipes A, E and I; and Table IV recipes E and M.

The incorporation of relatively small amounts of sulfur, for example Table II recipes B and F, yields vulcanizates which retain virtually completely their

original tensile strength on aging. For those stocks containing Philblack O (Table III) satisfactory cures are obtained with only 0.5 parts of triethylene tetramine. Tables II, III and IV show that high concentrations of triethylene tetramine should be avoided for the best retention of tensile properties and furthermore that high concentrations of sulfur apparently do not impart any additional resistance to heat.

It should be noted, however, that low concentrations of sulfur accelerate the vulcanization process. This is borne out by comparing recipes A and B in Table II. The addition of only 0.25 part of sulfur while the trienthylene tetramine concentration is constant at 0.75 part ethylene tetramine concentration is constant at 0.75 part increases the tensile strength from 790 to 1350 p.s.i. while the elongation decreases from 780 to 480%. By compromising between the best initial tensile properties and the best heat aged tensile properties several recipes were chosen for extensive evaluation. These recipes are: Table II G and J, Table III F and U, and Table IV B and K. This provided two recipes containing Furnex, two recipes containing Philblack O, and one recipe each for P-33 and Wyex blacks. Philblack O loaded stocks containing a minimum of curative, 0.5 part of triethylene tetramine in recipes B, C, and D in Table III, have

TABLE Sample G(II)* Cure Time (mins.) at 298°F 15 Tensile strength, p.s.i. 1430 Ultimate elongation, % 270 Shore A hardness (30 sec.) 54 Sample U(III) Cure Time (mins.) at 298°F 15 Tensile strength, p.s.i. 1560 Ultimate elongation, % 420 Ultimate elongation, % 360	G(II) 30 1500 270 340 58 U(III) 30 1750 390	G(II) 60 1500 250 370 60 U(III) 60 1790 310 480	G(II) 120 1540 200 600 62) U(III) 120 1840 270 590	15 1470 320 280 54	30 1470 260 380 56 B(IV) 30 1380 350 400	60 1470 260 360 60	120 1540 190 690 63	15 1690 380 380 60	F(III) 30 1790 330 430 65 K(IV) 30 1760 350 500 64	60 1670 280 530 69	F(III) 120 1620 220 720 71 K(IV) 120 1720 270 740 70
Modulus at 100%		68	74		these	- indicates	the table	where th	e recipe m	ay be loca	ited.

Shore A hardness (30 sec.).... or os section and the number in parentheses indicates the table where the recipe may be located.

* The letter indicates the recipe for preparing the vulcanizate and the number in parentheses indicates the table where the recipe may be located.

cutstanding heat resistance as exemplified by the fact that the vulcanizates still exhibited better than 450% elongation after aging for 72 hours at 350°F. Since many of the small laboratory batches of Lactoprene BN-12.5 did not yield satisfactory cures with this low concentration of triethylene tetramine it was decided to use higher amine concentrations in the evaluation study.

Table V shows the change in the initial tensile properties of the vulcanizates obtained with the selected recipes for curing times from 15 minutes to 120 minutes. These data show that for the various recipes 15 minutes at 298°F. results in vulcanizates which are sufficiently cured for many applications. In this study, therefore, the vulcanizates of Lactoprene BN-12.5 were cured for 15 minutes at 298°F.

Comparison with Other Synthetic Rubbers

For a proper comparison of the properties of Lactoprene BN-12.5 vulcanizates with other commercially available rubbers it was deemed advisable to actually evaluate the latter vulcanizates with those of Lactoprene BN. The following rubbers were included in this study: Paracril 18-80, Hycar OR-15, Hycar OR-25, Perbunan 26, and Hycar PA-21. The recipes and processing techniques employed were usually those recommended by the manufacturer for heat resistant applications. These recipes are presented in Table VI along with the initial tensile properties of the cured test slabs. These vulcanizates were subjected to the same experimental conditions and at the same time as the acrylic elastomer.

Resistance to Dry Heat

An intensive study of the resistance to dry heat of Lactoprene BN-12.5 vulcanizates was undertaken. The deterioration due to heat was studied in both an air-convection oven at 300° and 350°F. and in an enclosed system at 350° and 400°F. The change in the tensile properties such as tensile strength, elongation, modulus, and durometer hardness was observed over extended periods of time. Dumbbell samples cut from the test slabs were either hung in the oven or hung on racks in sealed containers. No attempt was made to remove or displace the residual air in the containers. Periodically

TABLE VI—RECIPES AND INITIAL TENSILE PROPERTIES OF COMMERCIAL RUBBERS USED IN THIS STUDY

Recipe Rubber	Paracril 18-80 100 1 54 2 3		Hycar OR-15 100 1 60 ——————————————————————————————	Perbunan 26 100 1 54 2 3	Hycar PA-21 100 1 50
	Initial	Propertie	:S		
Tensile strength, p.s.i Elongation, % Modulus at 100%, p.s.	2120 510 210	2550 260 740	2370 310 510	2140 550 200	1370 190 700
Shore A hardness (30 sec.)	. 60	75	71	63	69

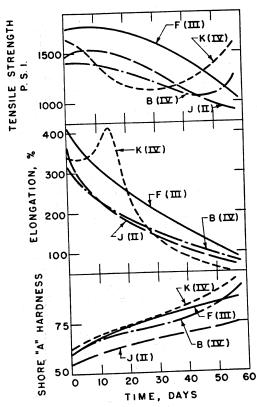


FIG. 1—Heat aging of Lactoprene BN-12.5 vulcanizates in an air-convection oven at 300°F.

samples were removed for testing. The commercial rubbers were evaluated similarly in the air convection oven at 300°F, and at 350°F, in the sealed containers.

Figure 1 shows the change of the tensile strength, elongation and hardness of the Lactoprene BN-12.5 vulcanizates, containing four different reinforcing agents, to dry heat at 300°F. for various times up to 56 days. Both the Furnex and Philblack O-loaded stocks show essentially a gradual lowering of tensile strength and elongation and a gradual increase in durometer hardness. At the end of 56 days at 300°F, the stocks still retained tensile strengths in excess of 900 p.s.i. and elongations better than 70%. The durometer hardness of the two stocks increased from an initial value of 55 and 60, respectively, to 75 and 85, respectively, at the end of 56 days' exposure. The P-33-loaded stocks show a similar decrease in tensile strength and elongation and a gradual increase in hardness. The Wyex stocks, however, showed a rather rapid decrease in tensile strength which remained approximately constant for about three weeks and then a rapid increase in tensile strength. The tensile strength of the Wyex stocks was initially 1640 p.s.i. and decreased to about 1120 p.s.i. at the end of 35 days; however, at the end of 56 days the tensile strength increased to 1760 p.s.i. The elongation of the Wyex stocks exhibited an initial elongation of 340% and increased to 410% in 14 days; subsequently, the elongation decreased most rapidly of the variously loaded stocks to a value of 50% at the end of the test. In addition, the Wyex stocks showed the greatest increase in hardness of the four reinforcing

agents studied.

The change in the physical properties of the other rubbers studied in the air-convection oven at 300°F. are shown in Figure 2. With the exception of Hycar PA-

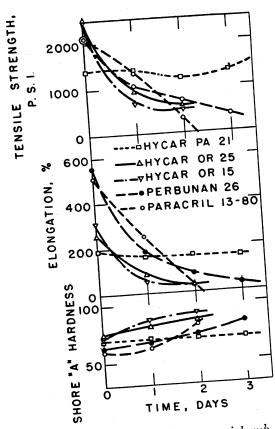


FIG. 2—Heat aging of commercial rubbers in an air-convection oven at 300°F.

21, all the samples had deteriorated at the end of four days to such an extent that they could not be tested.

At an even higher temperature, 350°F. in the air-convection oven, the Lactoprene BN-12.5 vulcanizates exhibited the change in physical properties shown in Figure 3. The Wyex-loaded vulcanizates showed the greatest decrease in tensile strength while the Furnex, Philblack O, and P-33-loaded vulcanizates showed very little change at the end of seven days. of these vulcanizates decreased to about 100% at the end of the exposure while the durometer hardness of the vulcanizates increased about 20 points above the initial

Since the exposure of the samples to heat in the airhardness values. convection oven does not approach the actual service conditions where these materials may find ultimate utility, such as gaskets, oil seals, and O-rings, the vulcanizates were placed in sealed containers and exposed for various lengths of time to temperatures of 350° and 400°F. It was felt that the above applications confine the samples of rubber so that air oxidation or the loss of oxidation inhibitors is minimized. Therefore, these conditions are more nearly simulated by exposing the samples to heat in an enclosed system.

Figure 4 illustrates the change in some of the tensile properties of the vulcanizates on exposure in sealed vessels to temperatures of 350°F. As can be seen, there is little change in either the tensile strength or elongation in the two differently reinforced stocks in 28 days while there is only a slight increase in the hardness values. There is materially less change in these tensile data than those obtained from samples exposed to 300°F. in the air-convection oven (see Figure 1).

Figure 5 illustrates the effect of 350°F. aging for

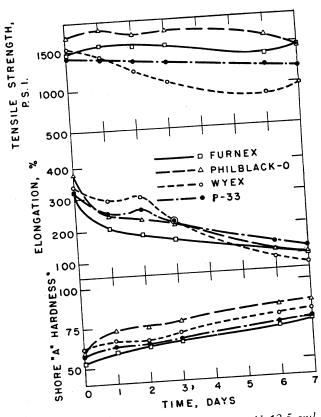


FIG. 3—Heat aging of Lactoprene BN-12.5 vulcanizates in an air-convection oven at 350°F.

the commercial rubbers under identical conditions. At the end of 15 days all the samples exhibited elongations of 50% and were in general so brittle that they could not be clamped in the tensile tester. The length of time to reach this state of deterioration was again consid-

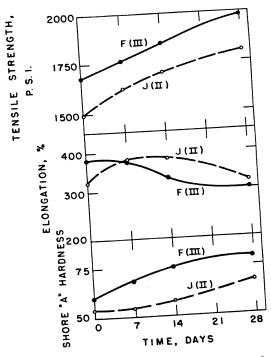


FIG. 4—Heat aging of Lactoprene BN-12.5 vulcanizates in sealed containers at 350°F.

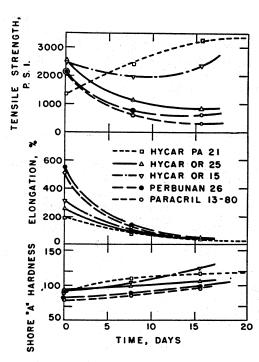
erably extended from that obtained at 300°F. in the air-convection oven (see Figure 2).

The effect of aging of the Lactoprene BN-12.5 vulcanizates at 400°F. in a sealed system is shown in Figure 6. The change in the tensile strength, elongation, and hardness are illustrated for samples exposed to 400°F. for various lengths of time in sealed vessels up to 28 days. For comparison, the data obtained for a silicone rubber vulcanizate is included. The Lactoprene BN stocks still retained tensile strengths better than 1300 p.s.i. and elongations somewhat greater than 150% after 28 days' exposure. The durometer hardness values increased about 20 points. The silicone rubber, designated G-E 12660 (kindly supplied by the General Electric Company), shows a much less percentage change in the tensile properties; however, its tensile strength and elongation are substantially lower than the acrylate-acrylonitrile vulcanizates.

Resistance to Hot Lubricating Oils

An intensive study of the oil resistance of Lactoprene BN-12.5 vulcanizates was also undertaken. The volume change as measured by the precent change in volume of rectangular specimens suspended in the ASTM Oils Numbers 1, 2 and 3 and a commercial automatic transmission oil at 300° and 350°F. was determined over a period of 28 days. In addition, the change in the tensile properties of dumbbells suspended in the hot oils was also determined. The latter oil, because of its exposure to extremely high temperatures, contains a number of additives to increase its useful life; unfortunately, many of these additives are extremely deleterious to most rubbers. This oil was included therefore to observe its effect on the Lactoprene BN vulcanizates.

It should be borne in mind that under the conditions of this test two effects are operating simultaneously. One is the effect of the elevated temperature upon the deterioration of the rubber which has been described in the previous section. The specimens immersed in the



IIG. 5—Heat aging of commercial rubbers in scaled containers at 350°F.

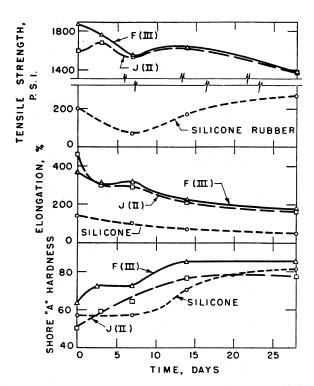


FIG. 6—Heat aging of Lactoprene BN-12.5 vulcanizates and a silicone rubber in an enclosed system at 400°F.

hot lubricating oils may be considered in a similar environment as those samples exposed to dry heat in the enclosed systems. However, the rate of chain scission and cross-linking in the oils, in comparison to these rates in the simple heat aging, is complicated by the additional action of the oil and its decomposition products on the vulcanizates. The change in tensile properties with immersion time in hot oils was used in this investigation as a criterion of the deterioration of the vulcanizates due to the heat and oil. The second effect is the volume swelling of the rubber in the hot oils. This is a thermodynamic property depending upon the number of cross-links and the polymer-oil interaction. It is beyond the scope of this paper to discuss the thermodynamic properties of this acrylic rubber.

Test dumbbells were prepared from the same test slabs as used in the heat aging studies. Rectangles 1-inch x 1½-inch cut from these slabs were used in the volume swelling determinations. This latter determination was performed following the procedure described in ASTM designation D471-46T. Furthermore, the oil was discarded every seven days and then replaced with fresh oil.

The percent volume increase and tensile data for four commercial rubbers are presented in Table VII. The samples were immersed in the lubricating oils up to 28 days at 300°F. The data in Table VII show some of the initial tensile properties of the vulcanizates and the tensile properties and volume swelling for two immersion times. At the end of 28 days the Lactoprene rubbers showed an increase in volume swelling of about 2% in ASTM Oil No. 1 between 9 and 12% in ASTM Oil No. 2, between 27 and 36% in ASTM Oil No. 3, and between 11 and 16% in the commercial automatic transmission fluid. Furthermore, the tensile strength and elongations of the acrylate vulcanizates were consistently high, showing negligible deterioration un-

Table VII—Volume Swelling and Tensile Properties of Vulcanizates Immersed in Various Oils at 300° F.

				Lenoth															•	
Sample				of Time												•	—Commercial	nercial A	utomatic	
Designation	Initia	1 Proper	-ties-	Immersed	1	STM	To. 1 Oil-		AA	STM No	No. 2 Oil-		Ķ	TM No.	3 Oil-		Tra	rsmission	Fluid	
Congination	T.S.	T.S. E. H.d	H.		T.S.	щi	H.	V.S.º			H.	v.S.		щi	Η	V.S. 1		щi	Ï	v.
J(II)*	1580	420	25	70 hours		230	23	4.0	1590	210	99	12.0		190	55	34.2		220 360	20 20 20 20 20 20 20 20 20 20 20 20 20	16.0 14.6
				28 days	1290	370	70	8.7	1230	€.~	48	0.21		220				3	3	2 0
F(III)	1690	430	61	70 hours	2040	300	85	2.8	1900	250 340	69 69	12.4 12.0	1260 960	250 340		25.5 1 36.4 1	620 400	270 360	61 61	16.0 15.2
B(IV)	1380	260	62	70 hours	1250	330	59	3.2	1140	420	99	11.2	850	430	555		1100	300	57	13.4
				28 days	1340	250	9	8.1	1100	350	3	2.6	00/	2/0			90	2	3	
K(IV)	1570	320	99	70 hours	1550	230	2%	3.8	1350	250	59.53	13.4	920 840	7 7 7 7 7 7 7 7 7 7 7 7 7	220	35.2 35.8		780 730 730	63	16.0 14.8
		1	1	co days	1400	027	2 ;	9 6	1470		3 1		750	130				RI. A.	×	
Hycar OR-25	2550	260	75	14 days 28 days	7380 1830 1830	750 150	4%	—2.5 —2.5	1470 560	38	25	3.1	480	100	218	14.2	720	100	71.	5.3
U DA 21	1270	160	9	14 days	1300	140	70	14	1580	150	9/	3.0	1370	180	29	13.3		1		1
11) (al 1 A-21	0/61	3	6	28 days	1510	110	. . .	-1.7	1540	130	29	2.4	1410	160		12.5	1470	140	76	2.0
Hycar OR-15	2470	310	71	14 days	2760	230	88	-2.2 -1.6	820	90 B L A	N K	1.2	610	100 B L A N	N K K	. 9.2	069	8	08	1.2
Perbunan 26	2140	550	63	14 days 28 days	2170 840	390 170	65.29	0.8	1810	350 130	54 55	13.2 12.6	1210 410	350 140	45 51	33.8	180	120	47	12.6
(a) Symbol denotes the table in paper and recipe for preparing vulcanizate.	s the table	in paper	and recipe	e for preparing v	vulcanizate.	(b) Tens	Tensile strength in lbs.	gth in Ibs	per sq.	in. (c) I	(c) Percent elongation.	ongation.	(d) Shore	(d) Shore A hardness (30		sec.). (e)	(e) Volume sv	swelling, %	•	

der these conditions. The commercial rubbers tested, with the exception of Hycar PA-21, exhibited a substantial decrease in tensile properties at the end of the test, particularly in the automatic transmission fluid.

Table VIII summarizes the volume swelling and tensile data obtained on the vulcanizates immersed in ASTM Oils Nos. 1 and 2 at 350°F. At this temperature, as above, the samples were immersed up to 28 days and data are shown for two immersion times. Only two Lactoprene vulcanizates were employed in this study at this temperature, sample numbers J (II) and F (III), loaded with Furnex and Philblack O, respectively. The Lactoprene rubbers showed very little deterioration at this temperature since both the tensile strengths and elongations remained fairly high up to 28 days. On the other hand, the elongations of the commercial rubbers (including Hycar PA-21) were extremely low at the end of the test.

Compression Set

Low compression set is extremely important for rubber vulcanizates to be employed in applications such as gaskets, O-rings, and similar mechanical goods. Since acrylate rubbers such as Lactoprene EV (Hycar PA-21) are being used in such applications, it is anticipated that these new acrylate-acrylonitrile rubbers should also provide material for similar mechanical specialties. Table IX summarizes the compression set data obtained on the four representative Lactoprene BN-12.5 vulcanizates and the four comparative rubbers tested at 250°F. for 70 hours.

The second column in the table shows the results obtained on the unaged vulcanizates. It shows that the compression set is inordinately high for the four Lactoprene BN-12.5 and Hycar PA-21 vulcanizates while two of the comparative rubber vulcanizates are low. However, since the Lactoprene rubbers have outstanding resistance to dry heat they may be conveniently aged or tempered in an air-convection oven for a short time prior to the compression set determination. This is common practice these days in the fabrication of certain specialty rubbers. This practice is unsatisfactory with most other rubbers because of their relatively poor heat resistance. The table shows that aging the vulcanizates substantially improves the compression set and that tempering for 24 hours at 350°F. in an air-convection oven is approximately the optimum condition, since longer exposure to this high temperature will affect the useful life of the rubber.

Table IX—Compression Set $^{\rm a}$ of the Lactoprene BN-12.5 Vulcanizates and Commercial Rubbers

Sample Designation	Initial Value	Tem- pered 2 hrs. at 300°F.	Tem- pered 2 hrs. at 350°F.	Tem- pered 24 hrs. at 300°F.	Tem- pered 24 hrs. at 350°F.
J(II) ^b F(III)	80.7 92.1	56.7 73.7	45.7	37.7 51.5	29.7 40.4
B(IV) K(IV)	77.6 86.8			49.5 59.5	47.9 54.3
Hycar OR-25. Hycar PA-21.	39.0 93.4		-	42.1	
Hycar OR-15. Perbunan 26	83.1 34.1				

⁽a) ASTM D-395-47-T, Method B. (b) Symbol denotes recipe and table for preparing vulcanizate.

Table X—Tensile Properties of Lactoprene BN-12.5 Vulcanizates and Commercial Rubbers

							Tensile	Propertie	s after 1	Aging 28	Davs in	Water
		•		Tens	sile Prope	rties		-	at 21	2°F.	•	
Sample	Initia	al Proper	ties	←Tem	pered 24	hrs.					pered 24	hrs -
Designation	Ur	itempered	1		at 300°F.		Ur	itempered			at 300°F.	
	T.S.b	E.°	H.d	T.S.	E.	H.	T.S.	Ē.	H.	T.S.	E.	H.
J(II) ^a	1600	460	50	1700	300	56	1020	130	62	800	170	58
F(III)	1880	370	63	1930	250	67	1330	130	72	800	160	60
B(IV)	1430	320	58	1380	260	62	1000	210	64	710	180	59
K(IV)	1640	340	62	1570	320	66	1020	100	68	710	140	59
Hycar OR-25	2550	260	75				2680	220	76			
Hycar PA-21	1370	190	69				380	100	54			
Hycar OR-15	2470	310	71				2600	210	79			
Perbunan 26	2140	550	63				2040	450	62			

⁽a) Symbol denotes recipe and table for preparing vulcanizates. (b) Tensile strength, p.s.i. (c) Ultimate elongation, %. (d) Shore A hardness (30 sec.).

of its high volume swelling, showed the greatest decrease in tensile properties.

Brittle Points

The most serious deficiency of Lactoprene EV vulcanizates originally developed at this Laboratory is the relatively high brittle point. This defect more than any other probably limits its utility since it restricts the number of applications where it may be employed. The use of butyl acrylate in place of ethyl acrylate as one of the co-monomers in the preparation of Lactoprene BN substantially lowers the brittle point of the vulcanizates.

Table XI summarizes the brittle point data obtained on the vulcanizates for a number of different exposure conditions as well as the initial brittle point of the unexposed samples. In addition, the brittle points of the comparative rubbers are included in the table. In the aging experiments of the Lactoprene samples at 300°F. in the air-convection oven, no significant change in the brittle points was observed in 72 hours. The Furnex-loaded stock had an initial brittle point of —11°F.; after 72 hours at 300°F., the observed brittle point was —15°F. At the end of four weeks, however, in the air-convection oven the brittle point was found to be +5°F. The Philblack O stock changed from an initial brittle point of —7°F. to +10.5°F. at the end of the four-week exposure.

The change in the brittle points of the comparative rubbers, with the exception of Hycar PA-21 which is an acrylic rubber and has a high brittle point initially, is much more pronounced. The Hycar OR-25 had a brittle point of -28.5°F. initially while at the end of 72 hours the brittle point increased to +23°F. The Hycar OR-15 had a brittle point initially of -4°F. which increased to above +36°F. in 24 hours. The Perbunan 26 rubber had initially a brittle point of -60°F. which increased to +8.5°F. in 72 hours. At 350°F. in the air-convection oven the Furnex and Philblack O stocks showed only a slight increase in the brittle point at the end of 72 hours. The brittle points of the samples aged in the various ASTM oils and the automatic transmission fluid showed a similar trend; however, the total change was not as great as in the air aging. In the oils as well the diene comparative rubbers exhibited the most pronounced change in brittle points.

Many specifications for rubber items to be used in the construction of mechanical goods require that the rubber have an extremely low brittle point. The brittle point is usually determined on the green stock which has not been exposed to any of the conditions which the rubber sample may encounter in service. These brittle point data suggest that a rubber with a slightly higher brittle point but with superior heat and oil resistance may be far superior in performance.

Effect of Storage of Compounded Stock

In order to expedite the evaluation of this new elastomer, the compounded stock was usually vulcanized about two hours after milling. This is not the recommended practice in rubber technology; usually, the milled stock is vulcanized 24 hours after compounding. Since this preferred technique was not followed, a study of the effect of storage on the compounded Lactoprene BN stock was not undertaken. The data in Table XII is typical of the data obtained from various storage experiments.

In these experiments, the butyl acrylate-acrylonitrile copolymer was generously supplied by the American Monomer Corporation and therefore this material was prepared under different experimental conditions from the copolymer used in the evaluation study. However, similar observations were observed with the copolymer prepared at the Government Evaluation Laboratory. The data presented in Table XII were obtained from vulcanizates prepared from the same recipe as F (III) with a curing time of 15 minutes at 300°F. for all the In all the experiments there appeared test samples. a significant lowering of the tensile strength and modulus in the first 24 hours. For the next three or four days the tensile properties remained fairly constant and then a gradual decrease of tensile properties would occur. However, no difficulty was encountered in removing the test slabs from the mold even at the end of the sixth day of storage. Storage under various humidity and

Table XII—Effect of Storage of the Compounded Stock on the Tensile Properties of Lactoprene BN-12.5 Vulcanizates

		——Ti	me in I	ays	
	0	1	2	3	6
Tensile strength, p.s.i Elongation, % Modulus at 300%, p.s.i Shore A hardness (30 sec.).	1350 640 800 48	1190 600 640 51	1200 630 650 46	1210 570 630 49	1010 690 540 48

Note: Copolymer prepared by American Monomer Corporation, Leominster, Mass. Recipe the same as F(III). 15 min. cure at 300°F.

Table XI—Brittle Points" (°F.) for Lactoprene BN-12.5 Vulcanizates and Commercial Rubbers for Various Exposure Conditions

Hycar OR-25

-23.0

(a) ASTM method of test D-746-44-T, manually operated.

(b) Symbol

indicates the table and recipe for preparing the vulcanizates,

B(IV)

J(II)

day

-15.0

9.5

-13.0

temperature conditions and remilling the stock before placing it in the mold did not appear to show any significant difference in the tensile data. Furthermore, variations in the amine-sulfur ratio in the recipe or temperature of milling resulted in a similar decrease of tensile strength and modulus in the first 24 hours and these properties would then remain fairly constant for the next few days.

Another paper in preparation will describe the properties of the vulcanizates obtained from the copolymer of ethyl acrylate-acrylonitrile and another copolymer of butyl acrylate-acrylonitrile. The latter, prepared from a monomer mixture of 95% butyl acrylate and 5% acrylonitrile, results in vulcanizates with lower brittle points but greater swelling in lubricating oils than Lactoprene BN-12.5 vulcanizates.

Summary

The data presented in this paper show that vulcanizates of Lactoprene BN-12.5, a copolymer of butyl acrylate and acrylonitrile, have certain properties superior to the vulcanizates of Lactoprene EV which was developed several years ago at this Laboratory. The Lactoprene BN vulcanizates have substantially lower volume swelling in water and lower brittle points than the vulcanizates of Lactoprene EV. Furthermore, the vulcanizates of this new rubber have outstanding resistance to dry heat, satisfactory oil resistance and low compression set. Because of these properties, Lactoprene BN-12.5 should provide improved materials for gaskets, O-rings, oil seals and similar items used in mechanical equipment.

ACKNOWLEDGMENT

The authors wish to acknowledge the cooperation of the Government Laboratories at the University of Akron in the preparation of the Lactoprene BN-12.5 sample used in the major part of this study. We are also indebted to the American Monomer Corporation for a sample of the copolymer; to the General Electric Company for a sample of a silicone vulcanizate; to T. J. Dietz for his helpful suggestions in the early stages of this development; to H. C. Fromuth for his assistance in obtaining some of the data presented in the paper, and to J. Siciliano for obtaining many of the brittle points.

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